

### Metropolis-Hastings MC: Nonsymmetric matrix $\alpha$ + 1/20 s12/4

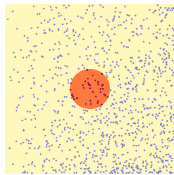
$\alpha_{i \rightarrow j}$  = matrix of the probability distribution of trial displacements

Metropolis:  $\alpha_{i \rightarrow j} = \alpha_{j \rightarrow i}$

What if  $\alpha_{i \rightarrow j} \neq \alpha_{j \rightarrow i}$ ?

$$W_{i \rightarrow j} = \begin{cases} \alpha_{i \rightarrow j} & \text{if } \pi_j \alpha_{j \rightarrow i} \geq \pi_i \alpha_{i \rightarrow j} \\ \alpha_{i \rightarrow j} \frac{\pi_j \alpha_{j \rightarrow i}}{\pi_i \alpha_{i \rightarrow j}} & \text{if } \pi_j \alpha_{j \rightarrow i} < \pi_i \alpha_{i \rightarrow j} \\ 1 - \sum_{k, k \neq i} W_{i \rightarrow k} & \text{for } i = j \end{cases}$$

$$p_{\text{acc}} = \min \left\{ 1, \frac{\alpha_{i \rightarrow j}}{\alpha_{j \rightarrow i}} \exp(-\beta \Delta U) \right\}$$



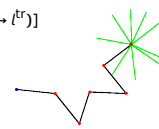
This extension of the Metropolis algorithm is attributed to W. K. Hastings

### Generalization for continuous models + 6/20 s12/4

- Instead of  $k$  possible direction on a lattice,  $k$  random trial configurations are generated
- The new configuration is drawn with the probability given by

$$p(i \rightarrow i^{\text{tr}}) = \frac{\exp[-\beta U(i \rightarrow i^{\text{tr}})]}{R_i}, \quad R_i = \sum_{l=1}^k \exp[-\beta U(i \rightarrow i^{\text{tr}})]$$

The configuration weight is:  $R = \prod_{i=1}^N R_i$



#### Limiting cases:

- For  $k = 1$ , the configuration weight is  $R = \exp[-\beta \sum U(i \rightarrow i^{\text{tr}})] = \exp(-\beta U_{\text{total}}^{\text{tr}}) =$  Boltzmann probability of a polymer. This is the naive MC integration.
- For  $k \rightarrow \infty$ , the method is close to the previous lattice-based method; i.e., drawing the new configuration with the Boltzmann probability at every step. (NB: this is not the importance sampling because the Rosenbluth weights differ.)

### Force bias + 2/20 s12/4

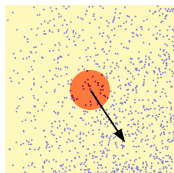
$$\alpha_{i \rightarrow j}^{\text{force bias}} = \alpha_{i \rightarrow j} \frac{\exp(\beta \lambda \vec{f}_k \cdot \Delta \vec{r})}{\int \exp(\beta \lambda \vec{f}_k \cdot \Delta \vec{r}) d\Delta \vec{r}}$$

Optimum  $\lambda = 0.5$  (between Metropolis and heat-bath)

Similarly:

- torque-bias (rotations)
- virial-bias (volume change)

$\lambda = 1$  & linearization  $\Rightarrow$  heat-bath



### More tricks +

- Global density change (close to the critical point)
- Tesselation to clusters and cluster moves
- NPT of hard bodies: molecules  $\rightarrow$  clusters, swell/shrink wrt cluster centers
- Swapping particles/molecules/groups/clusters

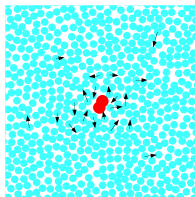
### Preferential sampling + 3/20 s12/4

Particles in the more interesting areas are moved more often.

Example: solvent molecules around a solute:  $p_{\text{pref}}(r) \stackrel{\text{e.g.}}{\approx} 1/(1+r^2/\sigma^2)$

#### Version 1:

- Choose a solvent molecule,  $i$ , at random
- If  $u_{[0,1]} < p_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)$  ( $\vec{r}_0$  = solute)
  - Generate a trial configuration  $\vec{r}_i^{\text{tr}}$
  - Accept it with  $p_{\text{acc}} = \min \left\{ 1, \frac{p_{\text{pref}}(|\vec{r}_i^{\text{tr}} - \vec{r}_0|)}{p_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)} \exp(-\beta \Delta U) \right\}$  (otherwise continue with the old configuration)



$$\alpha_{\text{pref}}(A \rightarrow A^{\text{tr}}) = \alpha(A \rightarrow A^{\text{tr}}) \frac{p_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)}{N}$$

### Configurational bias Monte Carlo + 7/20 s12/4

Instead of one trial configuration in the Metropolis method, let us consider  $k$  of them (continuous models: random sample | lattice models: all possible). A trial configurations is drawn from the probability distribution:

$$p(\vec{r}_i^{\text{tr}}) = \frac{\exp[-\beta U(\vec{r}_i^{\text{tr}})]}{R_{\text{tr}}}, \quad R_{\text{tr}} = \sum_{l=1}^k \exp[-\beta U(\vec{r}_l^{\text{tr}})]$$

It is accepted with probability  $\min \left\{ 1, \frac{R_{\text{tr}}}{R_{\text{old}}} \right\}$ , where

$$R_{\text{old}} = \exp[-\beta U(\vec{r}^{\text{old}})] + \sum_{l=2}^k \exp[-\beta U(\vec{r}_l^{\text{old}, \text{tr}})] \quad R_{\text{old}} = \sum_{l=1}^k \exp[-\beta U(\vec{r}_l^{\text{old}})]$$

( $\vec{r}_l^{\text{old}, \text{tr}}$  = random configuration generated in the same way)

NB: for lattices, it may happen  $R_{\text{old}} = R_{\text{tr}}$ , which is the heat bath method

Stochastic matrix

$$W_{i \rightarrow j} = \begin{cases} \alpha_{i \rightarrow j} \frac{\exp[-\beta U(\vec{r}_j)]}{R_j} & \text{for } R_j > R_i \\ \alpha_{i \rightarrow j} \frac{\exp[-\beta U(\vec{r}_j)] R_j}{R_i} & \text{for } R_j < R_i \end{cases}$$

### Simulations of molecular systems 8/20 s12/4

- Small molecules may be rigid (nitrogen, water, methane)
- Large molecules must be flexible... except (some) bonds

#### Vibrating (classical) bonds:

- simplicity and consistency of the model
  - technical problems with too stiff springs (short timestep in v MD, short trial moves in MC)
- code simplicity
- more realistic description of flexibility
  - vibrational frequencies (esp. for hydrogens) are so high that cannot be treated by classical mechanics anyway
  - transfer of energy between the fast vibrations and slow degrees of freedom is slow (they are decoupled), unless a stochastic thermostat is used
  - flexible models are more complicated theoretically

Fixed (constrained) bonds: just opposite

### Preferential sampling + 4/20 s12/4

#### Version 2:

- Choose a solvent molecule,  $i$ , from the distribution  $p_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)$ ,  $i = 1, \dots, N$   
An optimum algorithm with binary search has cost  $\propto \log N$ . Easy one:
  - select solvent  $i$  randomly
  - if not  $u_{[0,1]} < p_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)$ , new selection
- Perform one MC step using the molecule:
  - generate a trial configuration  $\vec{r}_i^{\text{tr}}$
  - accept it with probability

$$p_{\text{acc}} = \min \left\{ 1, \frac{p_{\text{pref}}(|\vec{r}_i^{\text{tr}} - \vec{r}_0|) S^{\text{tr}}}{p_{\text{pref}}(|\vec{r}_i - \vec{r}_0|) S} \exp(-\beta \Delta U) \right\}, \quad S = \sum_{i=1}^N p_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)$$

$$\alpha_{\text{pref}}(A \rightarrow A^{\text{tr}}) = \alpha(A \rightarrow A^{\text{tr}}) \frac{p_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)}{S}$$

Both versions are of the same efficiency if well optimized

### MC: molecules 9/20 s12/4

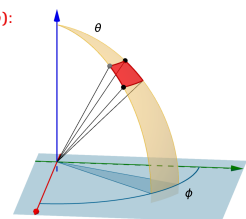
Example of a wrong algorithm for a linear molecule with axis  $(\theta, \phi)$ :



$$\begin{aligned} \phi^{\text{tr}} &= \theta + \Delta \theta u_{[-1,1]} \\ \theta^{\text{tr}} &= \phi + \Delta \phi u_{[-1,1]} \end{aligned}$$

Example of a correct algorithm for a general body:

- choose an axis randomly:
  - any of  $\hat{x}, \hat{y}, \hat{z}$  in the body frame
  - any of  $\hat{x}, \hat{y}, \hat{z}$  Cartesian coordinates in 3D space
  - any random vector



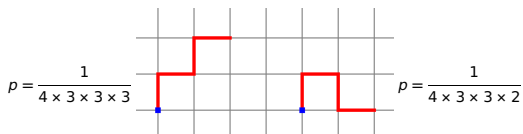
- rotate by angle  $\Delta \alpha u_{[-1,1]}$ , where  $u_{[-1,1]}$  is a random number uniformly distributed in  $[-1, 1]$

Rotation by  $\Delta \alpha$  around  $\hat{z}$ -axis:

$$\begin{pmatrix} \cos \Delta \alpha & -\sin \Delta \alpha & 0 \\ \sin \Delta \alpha & \cos \Delta \alpha & 0 \\ 0 & 0 & 1 \end{pmatrix}$$

### Polymers and Rosenbluth sampling 5/20 s12/4

A polymer in an athermal (very good) solvent is modeled by a self-avoiding random walk:



Both polymer should have the same weight!

Remedy: Rosenbluth weight (factor) of a step = number of possible continuations =  $R_i$ , the weight of a configuration obtained by the walk is:

$$R = \frac{1}{p} = \prod_{i=1}^N R_i$$

Generalization (the continuation is selected  $\propto$  Boltzmann factor):

$$R_i = \sum_{l=1}^k \exp[-\beta U(i \rightarrow l)]$$

### Quaternions + 10/20 s12/4

Quaternion:

$$q = w + xi + yj + zk$$

$$i^2 = j^2 = k^2 = -1, \quad ij = -ji = k, \quad ijk = -1 \text{ (a cykl.)}$$

$$|q|^2 = w^2 + x^2 + y^2 + z^2$$

Set  $\{|q| = 1, q \simeq -q\}$  is isomorphic with  $SO(3)$  (group of rotations), which can be used in MC, kinematics, and dynamics

Matrix of rotation:

$$\Omega = \begin{pmatrix} w^2 + x^2 - y^2 - z^2 & 2xy - 2zw & 2zx + 2yw \\ 2xy + 2zw & w^2 - x^2 + y^2 - z^2 & 2yz - 2xw \\ 2zx - 2yw & 2yz + 2xw & w^2 - x^2 - y^2 - z^2 \end{pmatrix}$$

There are 4 algebras (above  $\mathbb{R}$  or a field) with division, which are a vector space with a norm so that  $|xy| = |x||y|$ : real numbers, complex numbers, quaternions (multiplication is not commutative) and octonions (Cayley algebra, multiplication is not commutative nor associative).

Field is something with operations  $+$   $-$   $*$   $/$  with the same structure as  $\mathbb{R}$ .

Algebra is a vector space over a field (of "scalars") with " $\cdot$ ", where  $(x+y) \cdot z = x \cdot z + y \cdot z$ ,  $z \cdot (x+y) = z \cdot x + z \cdot y$ ,  $(ax) \cdot (by) = (ab)(x \cdot y)$  for vectors  $x, y, z$  and scalars  $a, b$ .


**MC: Molecules with internal degrees of freedom** 11/20  
s12/4

- Cartesian coordinates → generalized coordinates (Jacobian is needed – difficult)
- Cartesian coordinates → orthogonal coordinates (Jacobian does not change while moving along one coordinate)
- Frozen degrees of freedom (bonds)

**Polymers:**

- standard moves
- crankshaft move
- reptation for linear homopolymers: a tail bead is cut off and moved at the head
  - one random new position = standard Metropolis
  - more trial moves = configurational bias MC
- various enhance sampling methods: umbrella sampling, parallel tempering, Wang–Landau

<http://www.youtube.com/watch?v=rCTSG-SrShk&feature=related>



**Exercise: Conformations of butane** + 16/20  
s12/4

**Task:** Study the dihedral angle distribution of the dihedral angle in the CHARMM model of butane at an elevated temperature.

Try:

- flexible angles and bonds
- flexible angles, fixed bonds (SHAKE)
- fixed angles and bonds
- optionally with the dihedral potential and 1–4 interaction turned off

**Howtos:**

blend -h99 = angles with atoms heavier than 99 g/mol → equivalent bonds  
 cook -u9999 = bonds with  $K < 9999 \text{ K}\text{\AA}^2/\text{mol}$  will be flexible, other fixed

To turn all 1–4 interactions off one has to edit butane.blc:  
 – table dihedrals: write K = 0  
 – table sites erase \* in term #0 (interactions 1–4 → nothing)

**MD: bonds** 12/20  
s12/4

Integration of the equations of motion for systems with fixed bond lengths or angles is not the same as the infinitely-large force constant limit of the corresponding flexible system.

**What to fix:**

- nothing – short timestep, bad equipartition (convergence of different degrees of freedom)
- bonds with hydrogen only – longer timestep allowed, better (but not the best) equipartition  
 GROMACS: special algorithm for H,  $h = 4 \text{ ps}$
- all bond lengths – larger systematic errors, not good for rigid geometries (fullerene) and heavy atoms
- all bond lengths + angles with hydrogens – cheap and less precise, but longer timestep allowed + good equipartition
- all bond lengths + all angles – WRONG except small molecules

**Methods:**

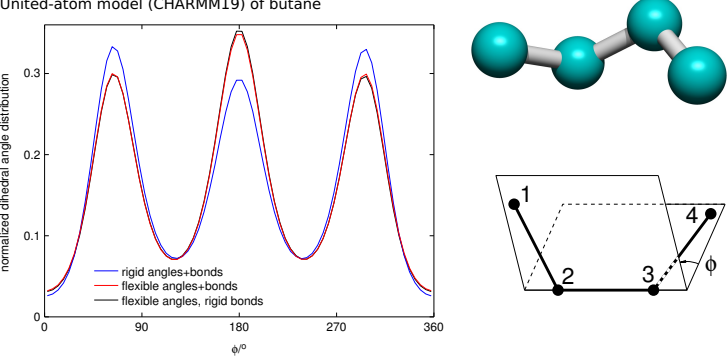
- SHAKE (+Verlet)
- Lagrangian constraint dynamics

**Exercise: Conformations of butane** + 17/20  
s12/4

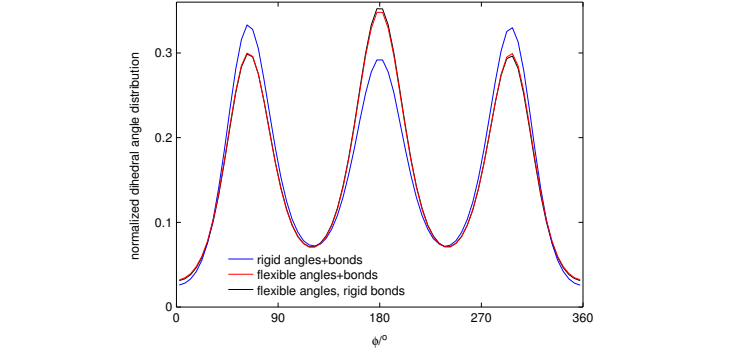
- unzip ../dihedral.zip
- flexible angles and bonds:  
`guest@403-a324-01:~/VY$ blend -o butane butane`  
`guest@403-a324-01:~/VY$ cookcedih butane dih -s -u9999`
- flexible angles, rigid bonds:  
`guest@403-a324-01:~/VY$ blend -o butane butane`  
`guest@403-a324-01:~/VY$ cookcedih butane dih -s`
- fixed angles and bonds:  
`guest@403-a324-01:~/VY$ blend -o butane -h99 butane`  
`guest@403-a324-01:~/VY$ cookcedih butane dih -s`

**Dihedral angle distribution of butane** [show/but1.sh] 13/20  
s12/4

United-atom model (CHARMM19) of butane

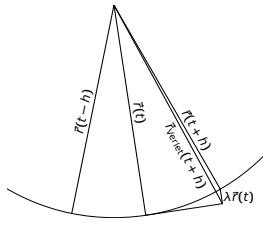


**Results** + 18/20  
s12/4



**Fixed constraints in MD: SHAKE** 14/20  
s12/4

**Mathematical pendulum:**

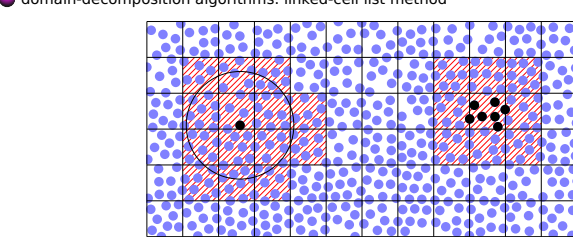


$$\begin{aligned} \vec{r}(t+h) &= \vec{r}_{\text{Verlet}}(t+h) - \frac{h^2}{m} \vec{f}_c(t) \\ &= 2\vec{r}(t) - \vec{r}(t-h) + h^2 \frac{\vec{f}(t) - \vec{f}_c(t)}{m} \\ \frac{h^2 \vec{f}_c(t)}{m} &= \lambda \vec{r}(t) \\ |\vec{r}(t+h)|^2 &= |\vec{r}(t)|^2 = l^2 \\ [\vec{r}_{\text{Verlet}}(t+h) - \lambda \vec{r}(t)]^2 &= \vec{r}(t)^2 \\ \vec{r}_{\text{Verlet}}(t+h)^2 - 2\lambda \vec{r}_{\text{Verlet}}(t+h) \cdot \vec{r}(t) + \lambda^2 \vec{r}(t)^2 &= \vec{r}(t)^2 \\ \lambda &\approx \frac{|\vec{r}_{\text{Verlet}}(t+h)|^2 - |\vec{r}(t)|^2}{2\vec{r}_{\text{Verlet}}(t+h) \cdot \vec{r}(t)} = \frac{|\vec{r}_{\text{Verlet}}(t+h)|^2 - l^2}{2\vec{r}_{\text{Verlet}}(t+h) \cdot \vec{r}(t)} \end{aligned}$$

**Optimization I** + 19/20  
s12/4

For short-ranged pair potentials (also r-space Ewald):

- all pairs ( $N < 300$ )
- neighbor list ( $N \approx 1000$ )
- domain-decomposition algorithms: linked-cell list method



**SHAKE** 15/20  
s12/4

**General atom-atom bond:**

$$\begin{aligned} \vec{r}_i(t+h) &= \vec{r}_{\text{Verlet},i}(t+h) + \lambda \frac{1/m_i}{1/m_i + 1/m_j} \vec{r}_{ij} \\ \vec{r}_j(t+h) &= \vec{r}_{\text{Verlet},j}(t+h) - \lambda \frac{1/m_j}{1/m_i + 1/m_j} \vec{r}_{ij} \end{aligned}$$

where

$$\lambda = \frac{|\vec{r}_{\text{Verlet},ij}(t+h)|^2 - |\vec{r}_{ij}(t)|^2}{2\vec{r}_{\text{Verlet},ij}(t+h) \cdot \vec{r}_{ij}(t)}$$

$\vec{r}_{ij} = \vec{r}_j - \vec{r}_i, r_{ij} = |\vec{r}_{ij}|$

- Based on the Verlet method
- Center-of-mass is conserved (integral of motion)!
- Complex molecules: repeat iteratively until converged
- Speed up: superrelaxation
- Velocity version: RATTLE, more variants
- For  $m_i = \infty, \vec{r}_i = (0, 0, 0), \vec{r} = \vec{r}_j = \vec{r}_{ij} \Rightarrow$  mathematical pendulum

**Optimization II** + 20/20  
s12/4

**A couple of tricks:**

- MD: multiple timestep MD
- MC: multimoves (near the critical point – critical slowing-down)
- MC: identity change, non-Boltzmann sampling, ...
- hybrid MC/MD (not very good)

**Programming tricks:**

- cache
- nearest neighbors in periodic boundary conditions
- tables: pair potential calculated by spines

**Parallel code:**

- usually based on domain decomposition (linked-cell list)
- standard computers 4–32 cores
- Graphics Processing Units: thousands of processors, more difficult to program