

Metropolis–Hastings MC: Nonsymmetric matrix α

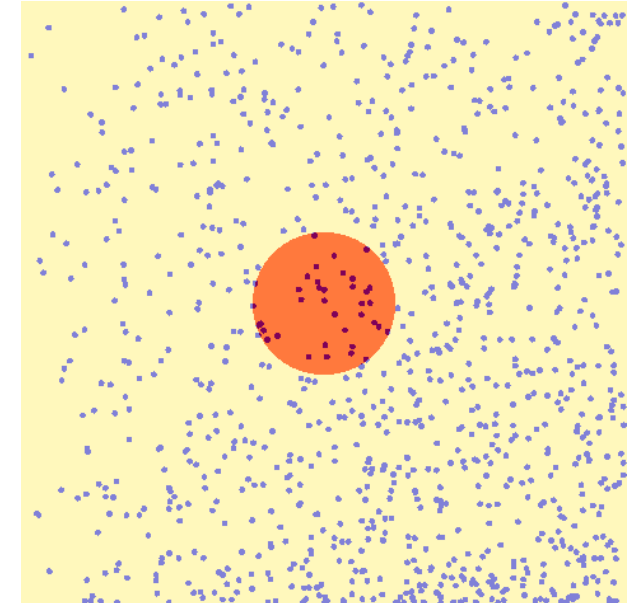
$\alpha_{i \rightarrow j}$ = matrix of the probability distribution of trial displacements

Metropolis: $\alpha_{i \rightarrow j} = \alpha_{j \rightarrow i}$

What if $\alpha_{i \rightarrow j} \neq \alpha_{j \rightarrow i}$?

$$W_{i \rightarrow j} = \begin{cases} \alpha_{i \rightarrow j} & \text{if } \pi_j \alpha_{j \rightarrow i} \geq \pi_i \alpha_{i \rightarrow j} \\ \alpha_{i \rightarrow j} \frac{\pi_j \alpha_{j \rightarrow i}}{\pi_i \alpha_{i \rightarrow j}} & \text{if } \pi_j \alpha_{j \rightarrow i} < \pi_i \alpha_{i \rightarrow j} \\ 1 - \sum_{k, k \neq i} W_{i \rightarrow k} & \text{for } i = j \end{cases}$$

$$p_{\text{acc}} = \min \left\{ 1, \frac{\alpha_{j \rightarrow i}}{\alpha_{i \rightarrow j}} \exp(-\beta \Delta U) \right\}$$



This extension of the Metropolis algorithm is attributed to W. K. Hastings

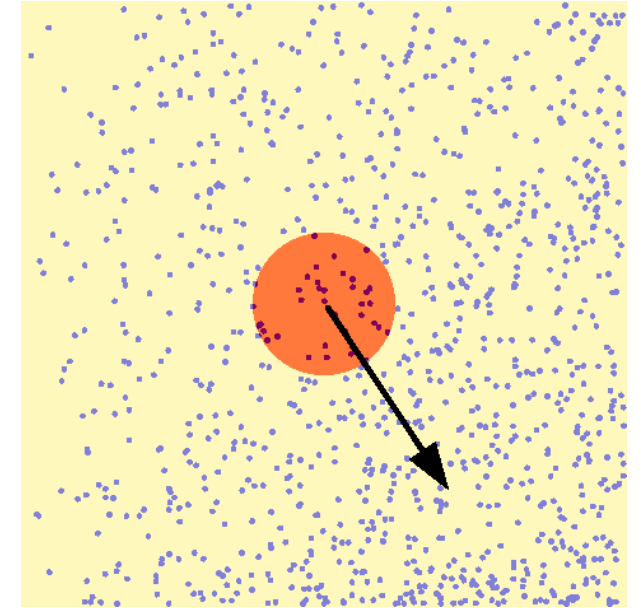
$$\alpha_{i \rightarrow j}^{\text{force bias}} = \alpha_{i \rightarrow j} \frac{\exp(\beta \lambda \vec{f}_k \cdot \Delta \vec{r})}{\int \exp(\beta \lambda \vec{f}_k \cdot \Delta \vec{r}) d\Delta \vec{r}}$$

Optimum $\lambda = 0.5$ (between Metropolis and heat-bath)

Similarly:

- torque-bias (rotations)
- virial-bias (volume change)

$\lambda = 1$ & linearization \Rightarrow heat-bath



More tricks



- Global density change (close to the critical point)
- Tessellation to clusters and cluster moves
- *NPT* of hard bodies: molecules \rightarrow clusters, swell//shrink wrt cluster centers
- Swapping particles/molecules/groups/clusters

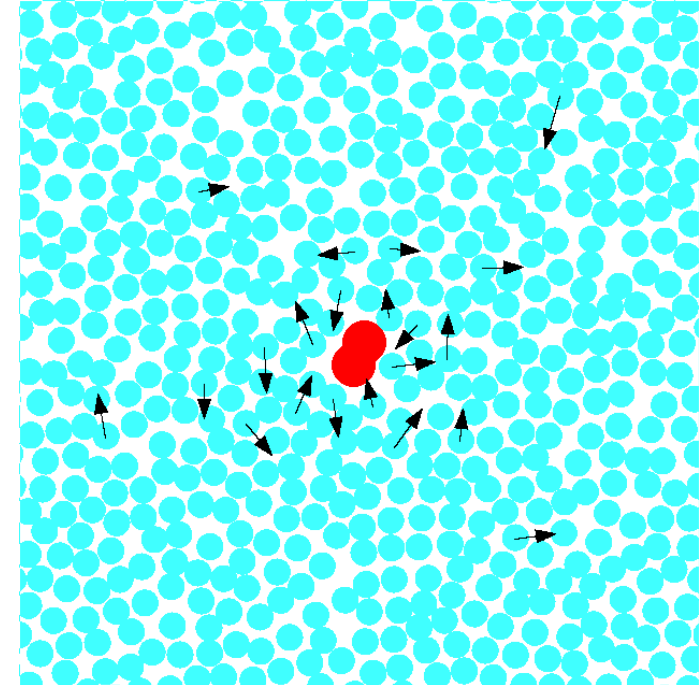
Particles in the more interesting areas are moved more often.

Example: solvent molecules around a solute: $\rho_{\text{pref}}(r) \stackrel{\text{e.g.}}{=} 1/(1 + r^2/\sigma^2)$

Version 1:

- Choose a solvent molecule, i , at random
- If $u_{[0,1)} < \rho_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)$ ($\vec{r}_0 = \text{solute}$)
 - Generate a trial configuration \vec{r}_i^{tr}
 - Accept it with $p_{\text{acc}} = \min \left\{ 1, \frac{\rho_{\text{pref}}(|\vec{r}_i^{\text{tr}} - \vec{r}_0|)}{\rho_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)} \exp(-\beta\Delta U) \right\}$
(otherwise continue with the old configuration)

$$\alpha_{\text{pref}}(A \rightarrow A^{\text{tr}}) = \alpha(A \rightarrow A^{\text{tr}}) \frac{\rho_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)}{N}$$



Version 2:

- Choose a solvent molecule, i , from the distribution $p_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)$, $i = 1, \dots, N$

An optimum algorithm with binary search has cost $\propto \log N$. Easy one:

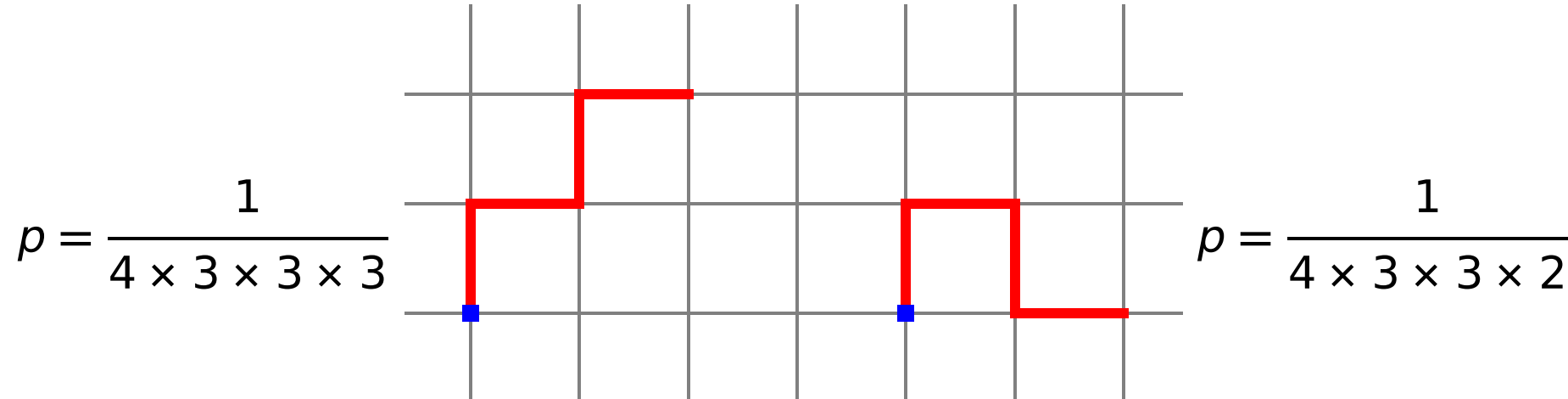
- select solvent i randomly
 - if not $u_{[0,1)} < p_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)$, new selection
- Perform one MC step using the molecule:
 - generate a trial configuration \vec{r}_i^{tr}
 - accept it with probability

$$p_{\text{acc}} = \min \left\{ 1, \frac{p_{\text{pref}}(|\vec{r}_i^{\text{tr}} - \vec{r}_0|)/S^{\text{tr}}}{p_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)/S} \exp(-\beta\Delta U) \right\}, \quad S = \sum_{i=1}^N p_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)$$

$$\alpha_{\text{pref}}(A \rightarrow A^{\text{tr}}) = \alpha(A \rightarrow A^{\text{tr}}) \frac{p_{\text{pref}}(|\vec{r}_i - \vec{r}_0|)}{S}$$

Both versions are of the same efficiency if well optimized

A polymer in an athermal (very good) solvent is modeled by a self-avoiding random walk:



Both polymer should have the same weight!

Remedy: Rosenbluth weight (factor) of a step = number of possible continuations = R_i , the weight of a configuration obtained by the walk is:

$$R = \frac{1}{p} = \prod_{i=1}^N R_i$$

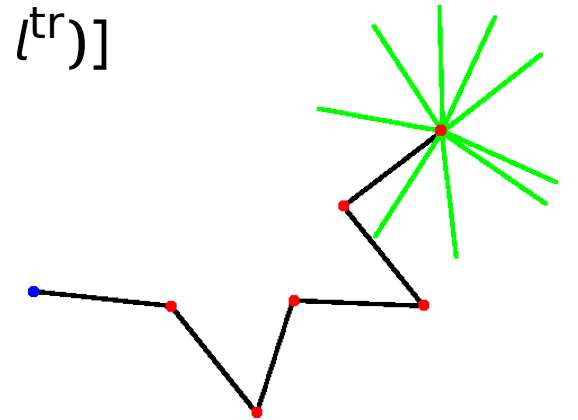
Generalization (the continuation is selected \propto Boltzmann factor):

$$R_i = \sum_{l=1}^k \exp[-\beta U(i \rightarrow l)]$$

- Instead of k possible direction on a lattice, k random trial configurations are generated
- The new configuration is drawn with the probability given by

$$p(i \rightarrow l^{\text{tr}}) = \frac{\exp[-\beta U(i \rightarrow l^{\text{tr}})]}{R_i}, \quad R_i = \sum_{l=1}^k \exp[-\beta U(i \rightarrow l^{\text{tr}})]$$

The configuration weight is: $R = \prod_{i=1}^N R_i$



Limiting cases:

- For $k = 1$, the configuration weight is $R = \exp[-\beta \sum U(i \rightarrow l^{\text{tr}})] = \exp(-\beta U_{\text{total}}^{\text{tr}}) =$ Boltzmann probability of a polymer. This is the naive MC integration.
- For $k \rightarrow \infty$, the method is close to the previous lattice-based method; i.e., drawing the new configuration with the Boltzmann probability at every step. (NB: this is not the importance sampling because the Rosenbluth weights differ.)

Configurational bias Monte Carlo

Instead of one trial configuration in the Metropolis method, let us consider k of them (**continuous models: random sample** | **lattice models: all possible**). A trial configurations is drawn from the probability distribution:

$$p(\vec{r}_l^{\text{tr}}) = \frac{\exp[-\beta U(\vec{r}_l^{\text{tr}})]}{R_{\text{tr}}}, \quad R_{\text{tr}} = \sum_{l=1}^k \exp[-\beta U(\vec{r}_l^{\text{tr}})]$$

It is accepted with probability $\min\left\{1, \frac{R_{\text{tr}}}{R_{\text{old}}}\right\}$, where

$$R_{\text{old}} = \exp[-\beta U(\vec{r}^{\text{old}})] + \sum_{l=2}^k \exp[-\beta U(\vec{r}_l^{\text{old, tr}})]$$

$$R_{\text{old}} = \sum_{l=1}^k \exp[-\beta U(\vec{r}_l^{\text{old}})]$$

($\vec{r}_l^{\text{old, tr}}$ = random configuration generated in the same way)

NB: for lattices, it may happen $R_{\text{old}} = R_{\text{tr}}$, which is the heat bath method

Stochastic matrix

$$W_{i \rightarrow j} = \begin{cases} \alpha_{i \rightarrow j} \frac{\exp[-\beta U(\vec{r}_j)]}{R_j} & \text{for } R_j > R_i \\ \alpha_{i \rightarrow j} \frac{\exp[-\beta U(\vec{r}_j)] R_j}{R_j R_i} & \text{for } R_j < R_i \end{cases}$$

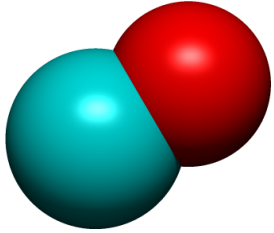
- Small molecules may be rigid (nitrogen, water, methane)
- Large molecules must be flexible ... except (some) bonds

Vibrating (classical) bonds:

- ⊕ simplicity and consistency of the model
- ⊕ code simplicity
- ⊕ more realistic description of flexibility
- ⊖ technical problems with too stiff springs (short timestep in v MD, short trial moves in MC)
- ⊖ vibrational frequencies (esp. for hydrogens) are so high that cannot be treated by classical mechanics anyway
- ⊖ transfer of energy between the fast vibrations and slow degrees of freedom is slow (they are decoupled), unless a stochastic thermostat is used
- ⊖ flexible models are more complicated theoretically

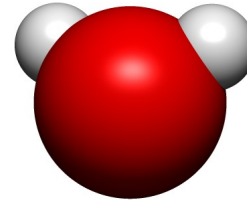
Fixed (constrained) bonds: just opposite

Example of a wrong algorithm for a linear molecule with axis (θ, ϕ) :



$$\begin{aligned}\theta^{\text{tr}} &= \theta + \Delta\theta u_{[-1,1]} \\ \phi^{\text{tr}} &= \phi + \Delta\phi u_{[-1,1]}\end{aligned}$$

Example of a correct algorithm for a general body:



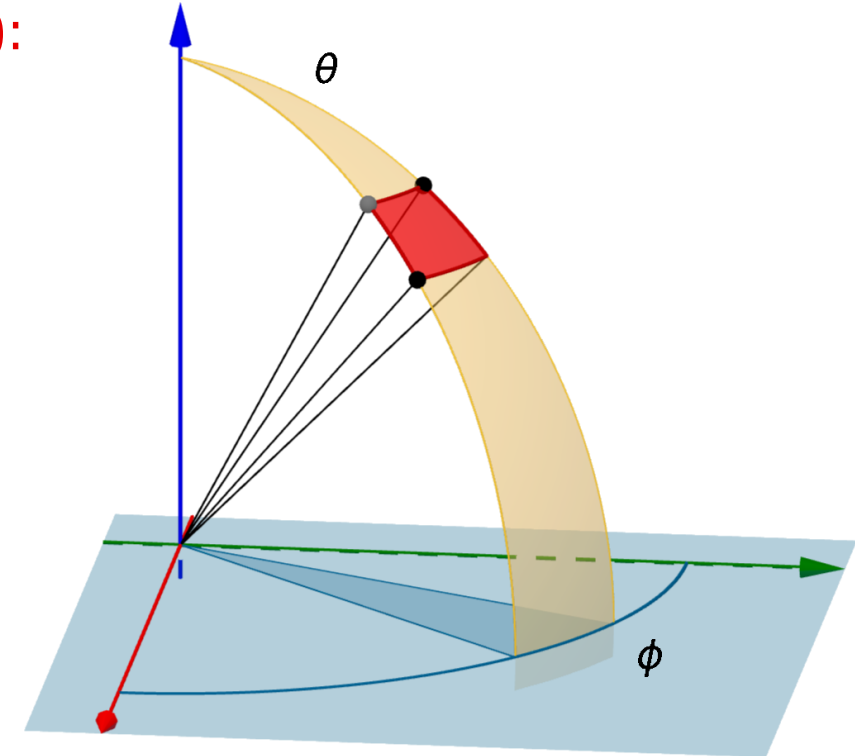
● choose an axis randomly:

- any of $\hat{x}, \hat{y}, \hat{z}$ in the body frame
- any of $\hat{x}, \hat{y}, \hat{z}$ Cartesian coordinates in 3D space
- any random vector

● rotate by angle $\Delta\alpha u_{[-1,1]}$, where $u_{[-1,1]}$ is a random number uniformly distributed in $[-1, 1]$

Rotation by $\Delta\alpha$ around \hat{z} -axis:

$$\begin{pmatrix} \cos \Delta\alpha & -\sin \Delta\alpha & 0 \\ \sin \Delta\alpha & \cos \Delta\alpha & 0 \\ 0 & 0 & 1 \end{pmatrix}$$



Quaternion:

$$q = w + xi + yj + zk$$

$$i^2 = j^2 = k^2 = -1, \quad ij = -ji = k, \quad ijk = -1 \text{ (a cykl.)}$$

$$|q|^2 = w^2 + x^2 + y^2 + z^2$$

Set $\{|q| = 1, q \simeq -q\}$ is isomorphic with $SO(3)$ (group of rotations), which can be used in MC, kinematics, and dynamics

Matrix of rotation:

$$\Omega = \begin{pmatrix} w^2 + x^2 - y^2 - z^2 & 2xy - 2zw & 2zx + 2yw \\ 2xy + 2zw & w^2 - x^2 + y^2 - z^2 & 2yz - 2xw \\ 2zx - 2yw & 2yz + 2xw & w^2 - x^2 - y^2 - z^2 \end{pmatrix}$$

There are 4 algebras (above \mathbb{R} or a field) with division, which are a vector space with a norm so that $|xy| = |x||y|$: real numbers, complex numbers, quaternions (multiplication is not commutative) and octonions (Cayley algebra, multiplication is not commutative nor associative).

Field is something with operations $+ - */$ with the same structure as \mathbb{R} .

Algebra is a vector space over a field (of "scalars") with " \cdot ", where $(x+y) \cdot z = x \cdot z + y \cdot z$, $z \cdot (x+y) = z \cdot x + z \cdot y$, $(ax) \cdot (by) = (ab)(x \cdot y)$ for vectors x, y, z and scalars a, b .

MC: Molecules with internal degrees of freedom

- Cartesian coordinates → generalized coordinates (Jacobian is needed – difficult)
- Cartesian coordinates → orthogonal coordinates (Jacobian does not change while moving along one coordinate)
- Frozen degrees of freedom (bonds)

Polymers:

- standard moves
- crankshaft move
- reptation for linear homopolymers: a tail bead is cut off and moved at the head
 - one random new position = standard Metropolis
 - more trial moves = configurational bias MC
- various enhance sampling methods:
umbrella sampling, parallel tempering, Wang–Landau

<http://www.youtube.com/watch?v=rCTSG-SrShk&feature=related>



Integration of the equations of motion for systems with fixed bond lengths or angles is not the same as the infinitely-large force constant limit of the corresponding flexible system.

What to fix:

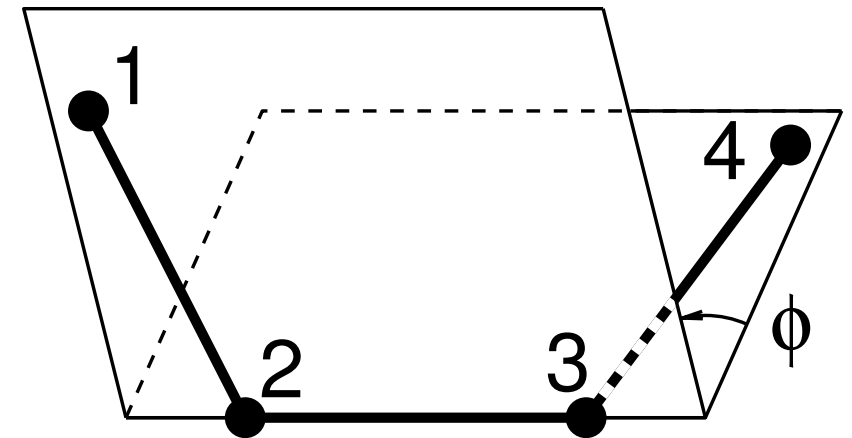
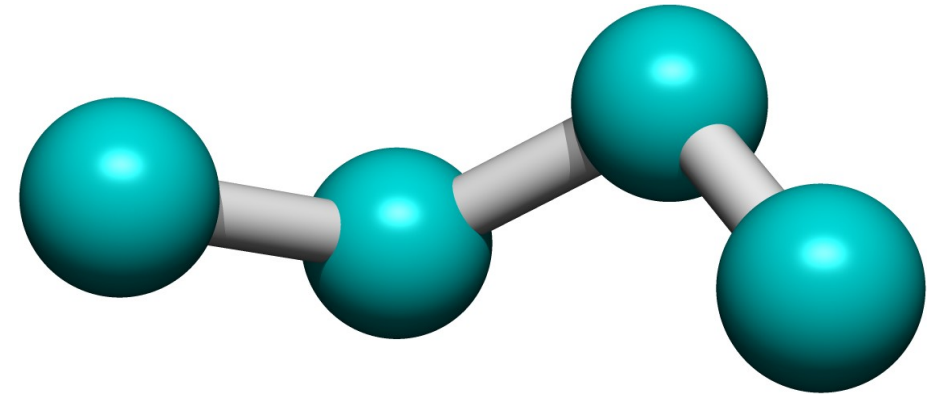
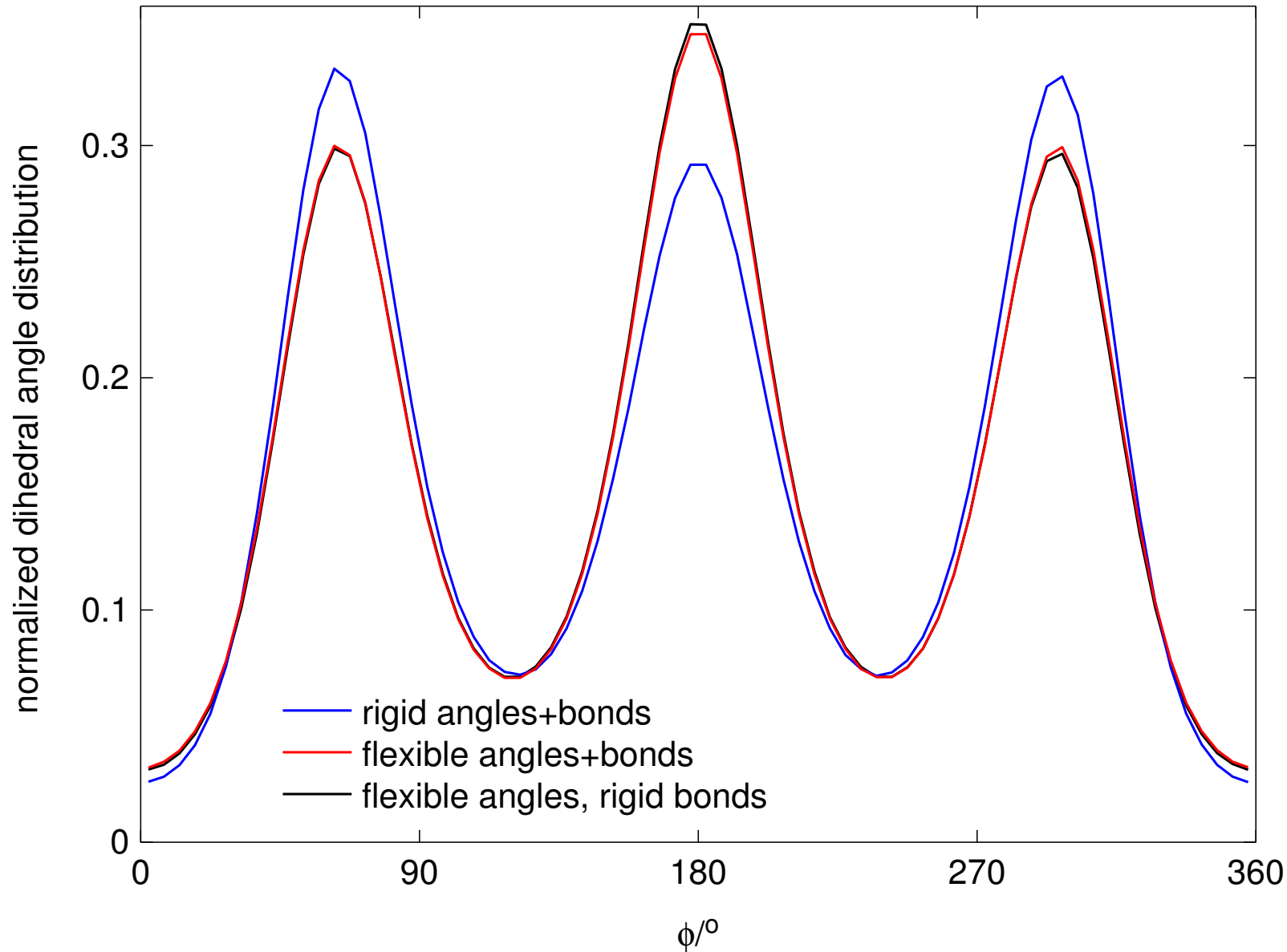
- nothing – short timestep, bad equipartition (convergence of different degrees of freedom)
- bonds with hydrogen only – longer timestep allowed, better (but not the best) equipartition
GROMACS: special algorithm for H, $h = 4$ ps
- all bond lengths – larger systematic errors, not good for rigid geometries (fullerene) and heavy atoms
- all bond lengths + angles with hydrogens – cheap and less precise, but longer timestep allowed + good equipartition
- all bond lengths + all angles – WRONG except small molecules

Methods:

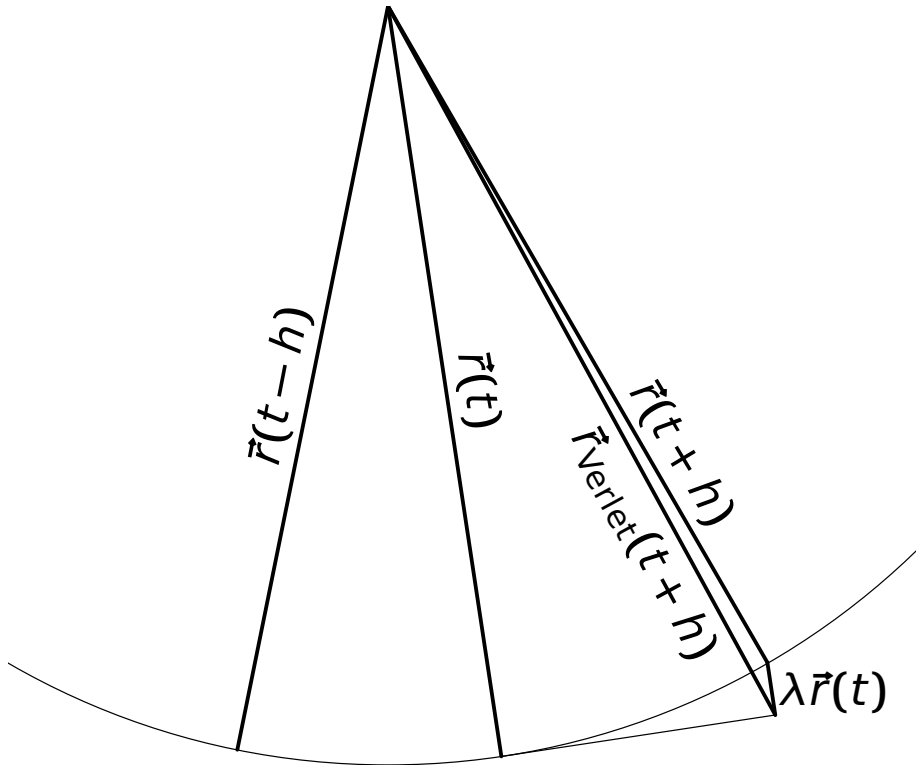
- SHAKE (+Verlet)
- Lagrangian constraint dynamics

Dihedral angle distribution of butane

United-atom model (CHARMM19) of butane



Mathematical pendulum:



$$\begin{aligned}\vec{r}(t+h) &= \vec{r}_{\text{Verlet}}(t+h) - \frac{h^2}{m} \vec{f}_c(t) \\ &= 2\vec{r}(t) - \vec{r}(t-h) + h^2 \frac{\vec{f}(t) - \vec{f}_c(t)}{m}\end{aligned}$$

$$\frac{h^2 \vec{f}_c(t)}{m} = \lambda \vec{r}(t)$$

$$|\vec{r}(t+h)|^2 = |\vec{r}(t)|^2 = l^2$$

$$[\vec{r}_{\text{Verlet}}(t+h) - \lambda \vec{r}(t)]^2 = \vec{r}(t)^2$$

$$\vec{r}_{\text{Verlet}}(t+h)^2 - 2\lambda \vec{r}_{\text{Verlet}}(t+h) \cdot \vec{r}(t) + \lambda^2 \vec{r}(t)^2 = \vec{r}(t)^2$$

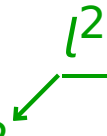
$$\lambda \approx \frac{|\vec{r}_{\text{Verlet}}(t+h)|^2 - |\vec{r}(t)|^2}{2\vec{r}_{\text{Verlet}}(t+h) \cdot \vec{r}(t)} = \frac{|\vec{r}_{\text{Verlet}}(t+h)|^2 - l^2}{2\vec{r}_{\text{Verlet}}(t+h) \cdot \vec{r}(t)}$$

General atom-atom bond:

$$\vec{r}_i(t+h) = \vec{r}_{\text{Verlet},i}(t+h) + \lambda \frac{1/m_i}{1/m_i + 1/m_j} \vec{r}_{ij}$$

$$\vec{r}_j(t+h) = \vec{r}_{\text{Verlet},j}(t+h) - \lambda \frac{1/m_j}{1/m_i + 1/m_j} \vec{r}_{ij}$$

where

$$\lambda = \frac{|\vec{r}_{\text{Verlet},ij}(t+h)|^2 - |\vec{r}_{ij}(t)|^2}{2\vec{r}_{\text{Verlet},ij}(t+h) \cdot \vec{r}_{ij}(t)}$$


- Based on the Verlet method
- Center-of-mass is conserved (integral of motion)!
- Complex molecules: repeat iteratively until converged
- Speed up: superrelaxation
- Velocity version: RATTLE, more variants
- For $m_i = \infty$, $\vec{r}_i = (0, 0, 0)$, $\vec{r} = \vec{r}_j = \vec{r}_{ij} \Rightarrow$ mathematical pendulum

$$\vec{r}_{ij} = \vec{r}_j - \vec{r}_i, \quad r_{ij} = |\vec{r}_{ij}|$$

Exercise: Conformations of butane

+ 16/20
s12/4

Task: Study the dihedral angle distribution of the dihedral angle in the CHARMM model of butane at an elevated temperature.

Try:

- flexible angles and bonds
- flexible angles, fixed bonds (SHAKE)
- fixed angles and bonds
- optionally with the dihedral potential and 1–4 interaction turned off

Howtos:

`bblend -h99` = angles with atoms heavier than 99 g/mol → equivalent bonds

`cook -u9999` = bonds with $K < 9999 \text{ K/\AA}^2/\text{mol}$ will be flexible,
other fixed

To turn all 1–4 interactions off one has to edit `butane.ble`:

- table `dihedrals`: write $K = 0$
- table `sites` erase * in term *0 (interactions 1–4 → nothing)

Exercise: Conformations of butane

+ 17/20
s12/4

● unzip ../dihedral.zip

● flexible angles and bonds:

```
guest@403-a324-01:~/VY$ blend -o butane butane
```

```
guest@403-a324-01:~/VY$ cookcedih butane dih -s -u9999
```

● flexible angles, rigid bonds:

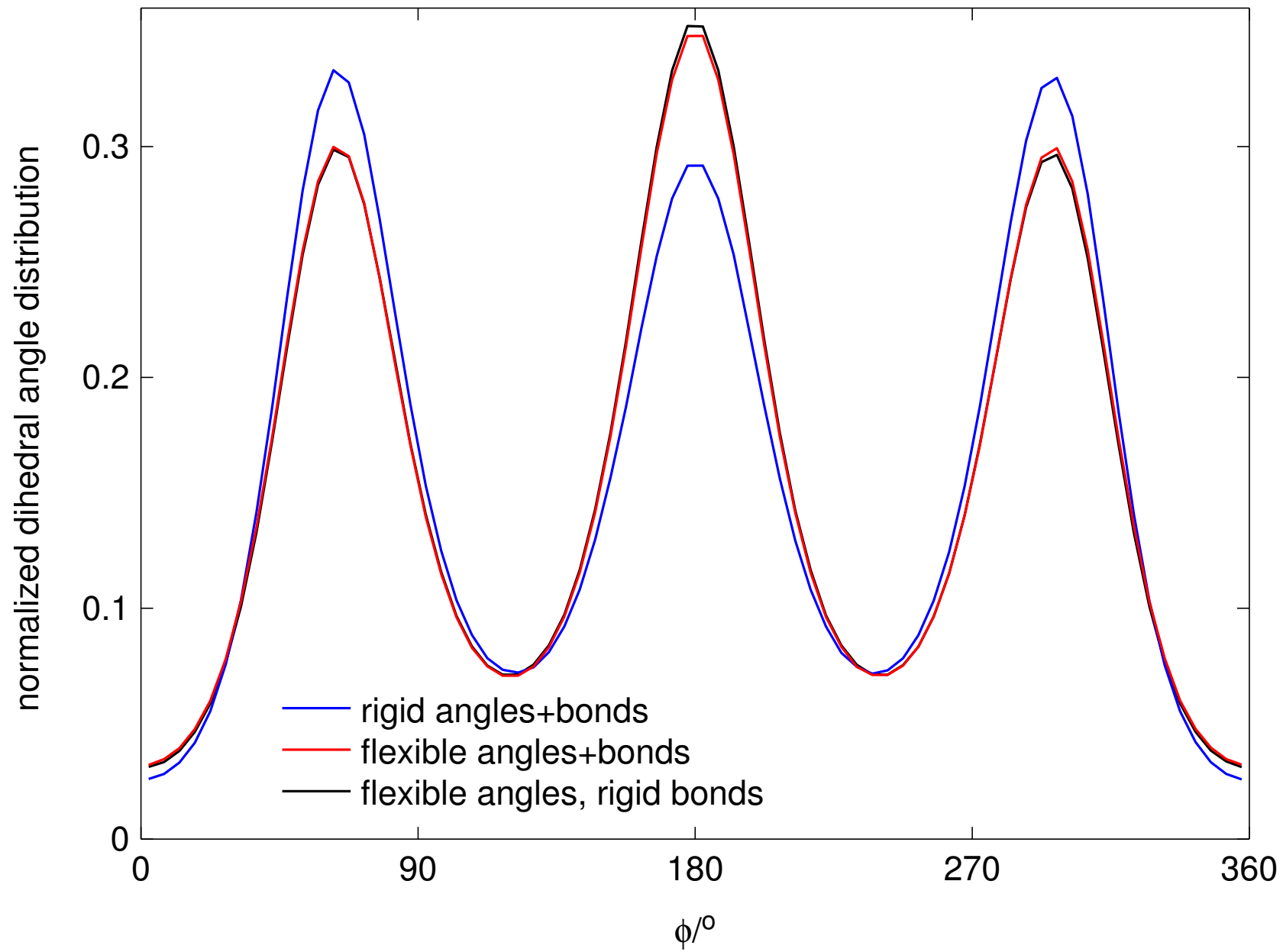
```
guest@403-a324-01:~/VY$ blend -o butane butane
```

```
guest@403-a324-01:~/VY$ cookcedih butane dih -s
```

● fixed angles and bonds:

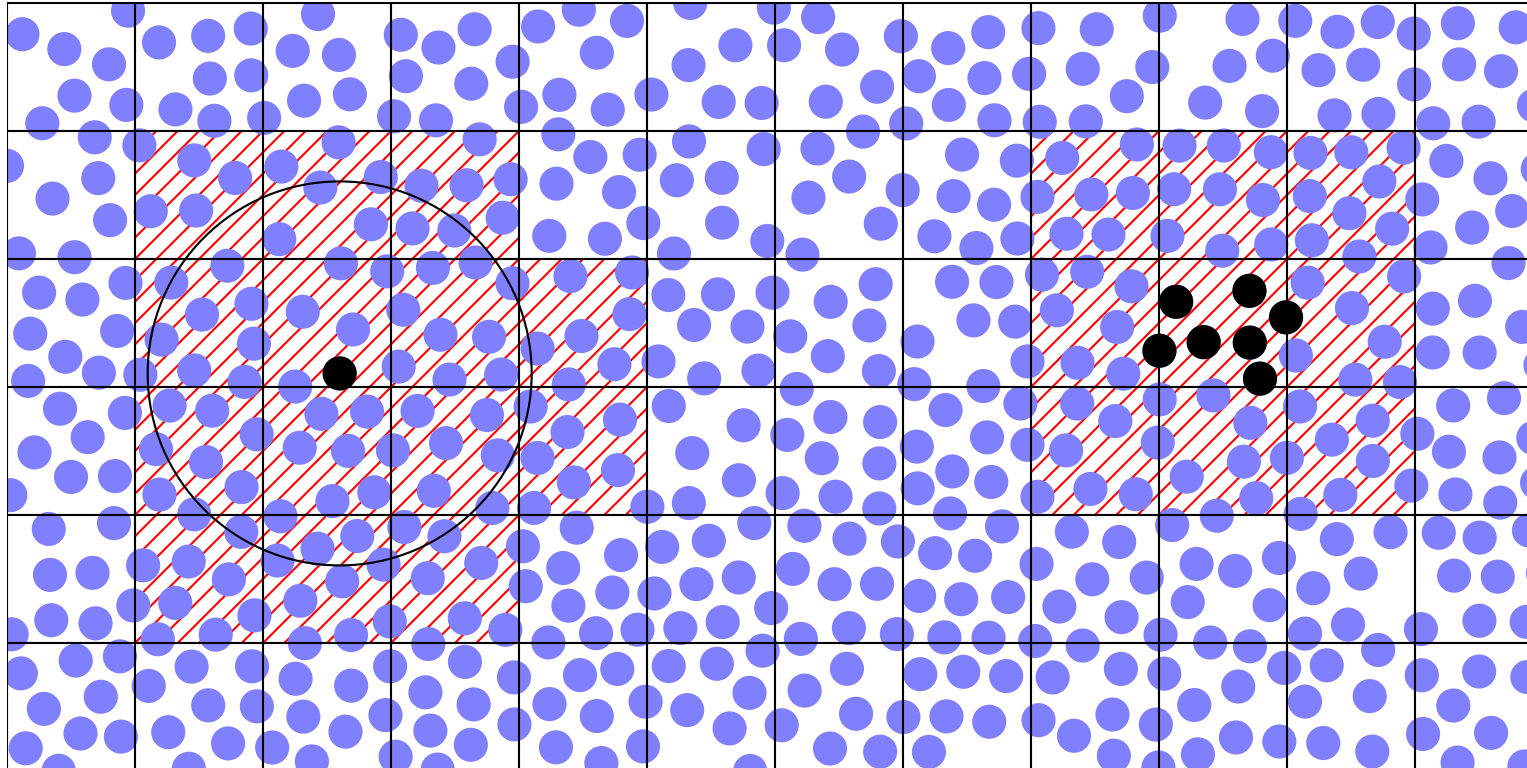
```
guest@403-a324-01:~/VY$ blend -o butane -h99 butane
```

```
guest@403-a324-01:~/VY$ cookcedih butane dih -s
```



For short-ranged pair potentials (also r-space Ewald):

- all pairs ($N < 300$)
- neighbor list ($N \approx 1000$)
- domain-decomposition algorithms: linked-cell list method



A couple of tricks:

- MD: multiple timestep MD
- MC: multimoves (near the critical point – critical slowing-down)
- MC: identity change, non-Boltzmann sampling, ...
- hybrid MC/MD (not very good)

Programming tricks:

- cache
- nearest neighbors in periodic boundary conditions
- tables: pair potential calculated by splines

Parallel code:

- usually based on domain decomposition (linked-cell list)
- standard computers 4–32 cores
- Graphics Processing Units: thousands of processors, more difficult to program